

1           **Supporting Information for “Kinematics of footwall**  
2           **exhumation at oceanic detachment faults: solid-block**  
3           **rotation and apparent unbending”**

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10          **Contents of this file**

11           1. Text S1 to S2

12           2. Figure S1

13           3. Table S1

14          **Additional Supporting Information (Files uploaded separately)**

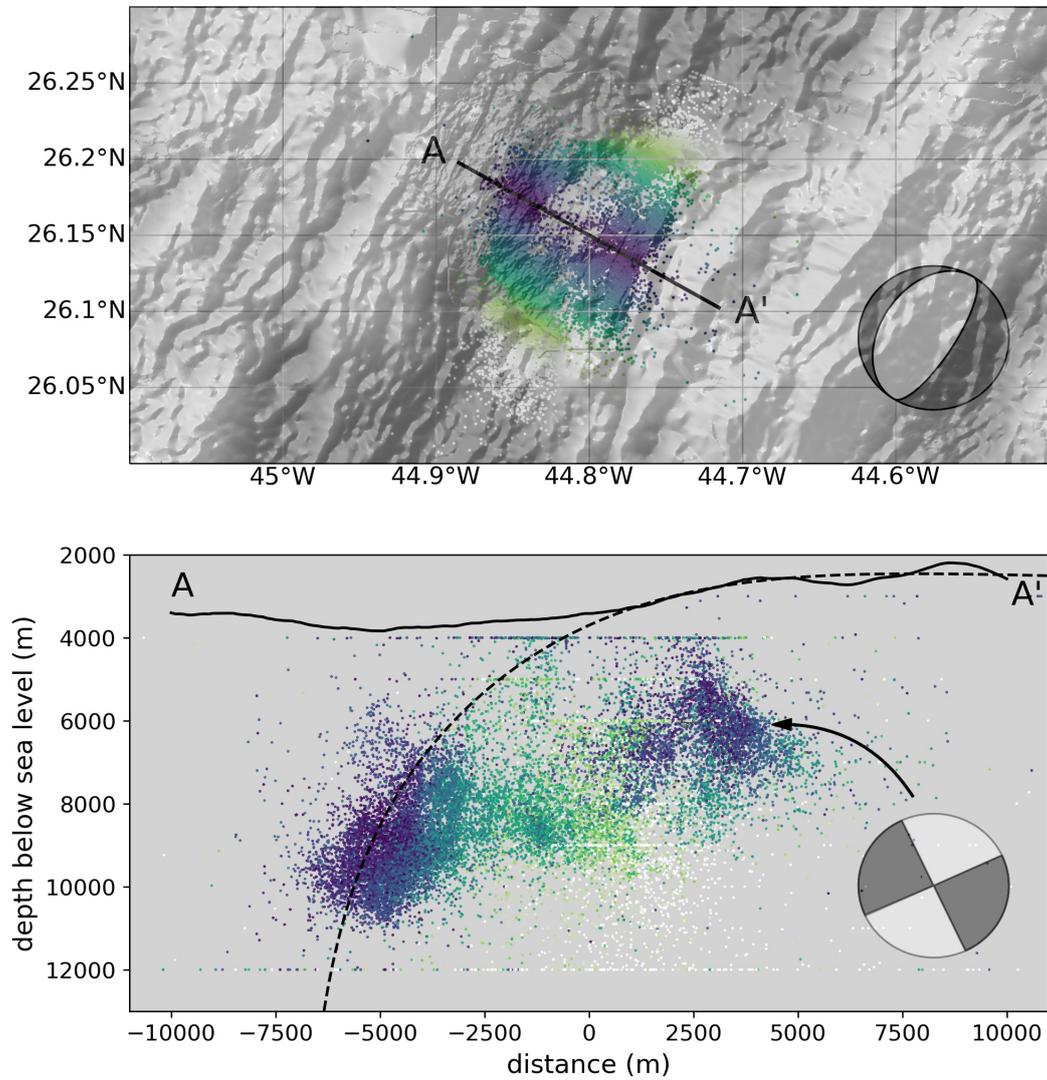
15           1. Captions for Movies S1 and S2

16           2. ASPECT parameter (.prm) file

17 **Text S1: TAG seismicity**

18 Fig. S1 shows map and cross-section views of the hypocenters from the Trans-Atlantic  
19 Geotraverse (TAG) detachment at the Mid-Atlantic Ridge, produced by Demartin et al.  
20 (2007). That study identified two distinct zones of seismic activity, one interpreted to  
21 represent the curved trace of the active detachment fault, and a second locus about 8  
22 km outboard of the detachment cluster, suggested to relate to antithetic normal fault  
23 planes in the footwall. It is notable that microseismicity along the detachment is con-  
24 centrated some 2-7 km beneath the seafloor. Craig and Parnell-Turner (2017) argue that  
25 the shallowest part of the TAG detachment (also the least-optimally oriented) tends to  
26 produce larger-magnitude earthquakes, and less microseismicity.

27 In map view, the TAG epicenters form a donut shape, with a prominent seismic  
28 gap in the footwall adjacent to the detachment fault termination at the seafloor. Focal  
29 mechanisms shown in Fig. S1 are representative, constructed from the dip values referred  
30 to in Demartin et al. (2007). Readers are referred to the original study for further de-  
31 tails. In the manuscript (Fig. 3) the earthquakes plotted are those within a distance  $\pm$   
32 4.5 km from the line A-A', which attempts to minimise the 3-D aspects of the full seis-  
33 micity pattern.



**Figure S1.** Hypocenters from the TAG detachment segment of the the Mid-Atlantic Ridge, from the study of Demartin et al. (2007). The distance of line between A-A' in the top panel is 20 km. Hypocenters are coloured to show relative distance from the line. Bottom panel shows a cross sectional view of the seismicity. The dashed line is a parameterization of the emergent detachment morphology from our numerical model.

34 **Text S2: Numerical model methods**35 **Thermo-mechanical model**

36 We model the 2-D thermo-mechanical evolution of an amagmatic oceanic spread-  
 37 ing center, using ASPECT to solve the incompressible Stokes and advection-diffusion equa-  
 38 tions, according to the Boussinesq approximations described in Bangerth et al. (2020b).  
 39 Adiabatic and shear heating are thus neglected in the energy equation. Elastic shear de-  
 40 formation is included in the constitutive model, necessitating an additional force term  
 41 in the Stokes equations (e.g. Schmalholz et al. (2001); Moresi et al. (2003); Bangerth et  
 42 al. (2020)). There is no compositional differentiation in the model (e.g. crust versus man-  
 43 tle) and the constitutive model applies to all parts of the domain. The temperature de-  
 44 pendence of the dislocation creep means that creep increasingly dominates at temper-  
 45 atures  $\geq 600^\circ\text{C}$ , while colder parts of lithosphere are effectively elasto-brittle. The con-  
 46 stitutive model is described in the following section.

47 **Viscous creep**

48 The effective viscosity associated with high temperature dislocation creep is mod-  
 49 elled with a wet olivine flow law (Hirth & Kohlstedt, 2003):

$$\eta = \frac{1}{2} A^{-\frac{1}{n}} |D|^{\frac{1-n}{n}} \exp\left(\frac{E + pV}{nRT}\right) \quad (1)$$

50  $|D|$  is the square root of the second invariant (or magnitude) of the deviatoric strain  
 51 rate tensor:  $|D| = (D_{ij}D_{ij}/2)^{1/2}$ .  $R$  is the gas constant,  $T$  is temperature,  $p$  is pres-  
 52 sure,  $A$  is the prefactor,  $n$  is the stress exponent,  $E$  is the activation energy and  $V$  is the  
 53 activation volume. Values are provided in Table S1.

54 The prefactor  $A$  is weakened linearly with accumulated viscous strain, following  
 55 the same functional form as the brittle strength weakening (e.g. Eqn. 11, see also Naliboff  
 56 et al. (2020)). Relevant parameters are given in Table S1.

57 **Visco-elasticity**

58 This section describes the implementation of Maxwell visco-elasticity within a Stokes  
 59 flow framework, where the stress history is tracked in an Eulerian reference frame (as

60 in ASPECT 2.2.0). Compared with a Lagrangian tracking scheme, such as described by  
 61 Moresi et al. (2003), the key difference is that advective terms must be accounted for in  
 62 the stress rate tensor.

63 In the Maxwell viscoelastic model, strain rates are proportional to the sum of the  
 64 stress and stress rate.  $D_{ij}$ , is given by

$$D_{ij} = D_{ij}^v + D_{ij}^e = \frac{\tau_{ij}}{2\eta} + \frac{1}{2\mu} \frac{D\tau_{ij}}{Dt}. \quad (2)$$

65 Where  $\tau_{ij}$  is the deviatoric part of the Cauchy stress tensor. To simplify the de-  
 66 scription in this section, we use  $\eta$  to refer to viscosity associated with dislocation creep  
 67 (i.e.  $\eta = \eta(T, p, |D|)$ ).

68 The constitutive relationship for a Maxwell viscoelastic fluid (Eqn. 2) contains the  
 69 stress rate tensor. The temporal derivative in the stress rate is a material derivative and,  
 70 as we will track the stress rate in a Eulerian reference frame, advective terms must be  
 71 accounted for.

72 A further requirement is that the stress rate tensor remains objective to rotation  
 73 experienced by the material parcels (see Schmalholz et al. (2001); R. J. Farrington (2017)  
 74 for details). This problem is typically handled by adopting an objective stress rate, in  
 75 order to enforce the objectivity. Following (Moresi et al., 2003), we employ the Zaremba-  
 76 Jaumann definition of stress rate:

$$\frac{D\tau_{ij}}{Dt} = \frac{\partial\tau_{ij}}{\partial t} + v_k \frac{\partial\tau_{ij}}{\partial x_k} - W_{ik}\tau_{kj} + \tau_{ik}W_{kj} \quad (3)$$

78 where  $W$  is the spin tensor.

79 Following Schmalholz et al. (2001); Moresi et al. (2003), ASPECT 2.2.0 discretizes  
 80 the temporal part of  $\frac{D\tau_{ij}}{Dt}$  using backwards finite difference:

$$\frac{\partial\tau_{ij}^t}{\partial t} \approx \frac{\tau_{ij}^t - \tau_{ij}^{t-\Delta t}}{\Delta t}$$

(4)

 81 Solving for  $\tau^t$ :

$$\begin{aligned}
 D_{ij}^t &= \frac{1}{2\eta}\tau_{ij}^t + \frac{1}{2\mu}\left(\frac{\tau_{ij}^t}{\Delta t} - \frac{\tau_{ij}^{t-\Delta t}}{\Delta t} + v_k \frac{\partial \tau_{ij}^{t-\Delta t}}{\partial x_k} - W_{ik}\tau_{kj}^{t-\Delta t} + \tau_{ik}^{t-\Delta t}W_{kj}\right) \\
 \left(\frac{\mu\Delta t}{\mu\Delta t\eta} + \frac{\eta}{\mu\Delta t\eta}\right)\tau_{ij}^t &= 2D_{ij}^t + \frac{1}{\mu}\left(\frac{\tau_{ij}^{t-\Delta t}}{\Delta t} + v_k \frac{\partial \tau_{ij}^{t-\Delta t}}{\partial x_k} - W_{ik}\tau_{kj}^{t-\Delta t} + \tau_{ik}^{t-\Delta t}W_{kj}\right) \\
 \tau_{ij}^t &= \left(\frac{\mu\Delta t\eta}{\mu\Delta t + \eta}\right)\left(2D_{ij}^t + \frac{1}{\mu}\left(\frac{\tau_{ij}^{t-\Delta t}}{\Delta t} + v_k \frac{\partial \tau_{ij}^{t-\Delta t}}{\partial x_k} - W_{ik}\tau_{kj}^{t-\Delta t} + \tau_{ik}^{t-\Delta t}W_{kj}\right)\right) \\
 \tau_{ij}^t &= \eta_{\text{eff}}\left(2D_{ij}^t + \frac{1}{\mu}\left(\frac{\tau_{ij}^{t-\Delta t}}{\Delta t} + v_k \frac{\partial \tau_{ij}^{t-\Delta t}}{\partial x_k} - W_{ik}\tau_{kj}^{t-\Delta t} + \tau_{ik}^{t-\Delta t}W_{kj}\right)\right)
 \end{aligned}$$

 82 For brevity, define  $\tilde{\tau}_{ij}$  as the stress history tensor advected and rotated into the con-  
 83 figuration of the current timestep:

$$\tilde{\tau}_{ij} = \left(\tau_{ij}^{t-\Delta t} + v_k \frac{\partial \tau_{ij}^{t-\Delta t}}{\partial x_k} - W_{ik}\tau_{kj}^{t-\Delta t} + \tau_{ik}^{t-\Delta t}W_{kj}\right) \quad (5)$$

84 so that

 85 The stress at timestep  $t$  is given by:

$$\tau^t = \eta_{\text{eff}}\left(2D_{ij}^t + \frac{1}{\mu\Delta t}\tilde{\tau}_{ij}\right) \quad (6)$$

 86 The Stokes Equation, representing conservation of momentum at infinite Prandtl  
 87 number, can then be modified as follows:

$$(2\eta_{\text{eff}}D_{ij})_{,j} - p_{,i} = f_i - \frac{\eta_{\text{eff}}}{\mu\Delta t}\tilde{\tau}_{ij,j} \quad (7)$$

 88 ***Advection and rotation terms in the stress rate***

 89 In ASPECT v2.2.0 (Bangerth et al., 2020), the stress history tensor is stored (com-  
 90 ponent wise) as a set of non-diffusive scalar compositional fields. In the current imple-

91 mentation a two-stage approach is used to approximate the Zaremba-Jaumann stress rate.  
 92 The advection terms for each component of stress rate are handled by the ASPECT's  
 93 default compositional field capability. Version 2.2.0 of ASPECT uses a 2nd order implicit  
 94 time integration for the advection equations (BDF-2).

95 Whenever the components of the stress history tensor are accessed (e.g. by var-  
 96 ious ASPECT material models) the relevant advection terms for each component will  
 97 already have been calculated. The rotation terms in the Zaremba-Jaumann stress rate  
 98 are then applied in the 'elasticity' submodule

99 (`aspect/source/material_model/rheology/elasticity.cc`):

$$\tilde{\tau}_{ij}^t = \frac{1}{\mu} \left( \frac{\tilde{\tau}_{ij}^t}{\Delta t} - W_{ik} \tilde{\tau}_{kj}^t + \tilde{\tau}_{ik}^t W_{kj} \right) \quad (8)$$

100 Where  $\tilde{\tau}_{ij}^t$  refers to the stress history tensor after advective terms have been han-  
 101 dled.

102 At the completion of the Stokes solve and the progression to the next time step,  
 103 the components of the stress history tensor need to be updated. This process is also han-  
 104 dled using ASPECT's compositional field capability. The update increment to the stress  
 105 history components are applied as a 'reaction term', i.e. a source term in the advection  
 106 equation.

107 Following R. Farrington et al. (2014), instead of simply taking the stress history  
 108 at  $t - 1$ , we store the stress history term  $\tilde{\tau}$  as a running average ( $\bar{\tau}_{ij}$ ) defined as:

$$\bar{\tau}_{ij} = (1 - \Phi) \tilde{\tau}_{ij} + \Phi \tau_{ij} \quad (9)$$

109 where  $\Phi = \Delta t_c / \Delta t_e < 1$ .

### 110 **Visco-elasto-plastic model**

Plastic deformation is incorporated into the visco-elastic constitutive model, fol-  
 lowing Moresi et al. (2003). Brittle behaviour is modelled through a Drucker-Prager yield  
 limit ( $\tau_y$ ) on the magnitude of the deviatoric stress:

$$\tau_y = p \sin(\phi) + C \cos(\phi) \quad (10)$$

111 where  $p$  is the pressure. The cohesion ( $C$ ) and friction angle ( $\phi$ ) are weakened with  
 112 accumulated plastic strain ( $\gamma^p$ ) according to:

$$C(\beta) = \beta C_1 + (1 - \beta) C_0 \quad (11)$$

113 Where

$$\beta = \min(1, \gamma^p / \gamma_0^c) \quad (12)$$

114 The model is initialised with plastic strain on the quadrature points, randomly sam-  
 115 pled from a uniform distribution between 0 and 0.25.

116 Again we use the notation  $\tilde{\tau}$  (omitting component indexes here for brevity) for the  
 117 stress history tensor (advected and rotated into the configuration of the current timestep).  
 118 Define an effective strain rate as:

$$D_{\text{eff}} = 2D + \frac{1}{\mu \Delta t_e} \tilde{\tau} \quad (13)$$

119 with the magnitude given by:  $|D| = (D_{ij} D_{ij} / 2)^{1/2}$ . The plastic effective viscos-  
 120 ity is then defined as:

$$\eta_p = \frac{\tau_y}{|D_{\text{eff}}|} \quad (14)$$

121 Substituting (14) into the definition of the stress (Eqn. 6) shows that this defini-  
 122 tion of the plastic viscosity satisfies the yield stress (i.e. it produces the intended viscos-  
 123 ity rescaling at each iteration).

124 The final viscosity  $\eta_{\text{vep}}$  is defined depending on whether the magnitude of the de-  
 125 viatoric stress tensor exceeds  $\tau_y^t$ :

$$\eta_{\text{vep}} = \begin{cases} \eta_p, & |\tau^t| \geq \tau_y \\ \eta_{\text{eff}}, & \text{otherwise} \end{cases}$$

126 A successive substitution (Picard) approach is used to resolve the nonlinearity in  
 127 the material model. The maximum number of iterations is limited to 40.

## **Model parameters**

Parameter name	Value	Symbol	Units
Model domain depth	100	-	km
Model domain width	400	-	km
Potential temperature	1573	$T_p$	K
Surface temperature	293	$T_s$	K
Viscosity minimum	$1 \times 10^{18}$	-	Pa s
Viscosity maximum	$1 \times 10^{24}$	-	Pa s
Dislocation creep volume	$22 \times 10^{-6}$	V	$\text{m}^3 \text{mol}^{-1}$
Dislocation creep energy	520	E	$\text{kJ mol}^{-1}$
Dislocation creep exponent	3.5	n	-
Initial dislocation creep prefactor	$3.77 \times 10^{-14}$	$A_0$	$\text{Pa}^{-n} \text{s}^{-1}$
Weakened dislocation creep prefactor	$1.385 \times 10^{-14}$	$A_1$	$\text{Pa}^{-n} \text{s}^{-1}$
Prefactor weakening interval	2	$\gamma_0^A$	-
Initial friction angle	30	$\phi_0$	$^\circ$
Initial cohesion	20	$C_0$	MPa
Weakened friction angle	3	$\phi_1$	-
Weakened cohesion	10	$C_1$	MPa
Friction angle weakening interval	6	$\gamma_0^\phi$	-
Cohesion weakening interval	6	$\gamma_0^C$	-
Elastic shear moduli	10	$\mu$	GPa
Thermal diffusivity	$3 \times 10^{-6}$	-	$\text{km}^2 \text{s}^{-1}$
Heat capacity	1000	$C_p$	$\text{JK}^{-1} \text{kg}^{-1}$
Full spreading rate	2	-	$\text{cm yr}^{-1}$
Elastic timestep	$10^4$	$\Delta t_e$	yr
Numerical timestep (max)	$2 \times 10^3$	$\Delta t_c$	yr
Reference density	3300	$\rho_0$	$\text{kg m}^{-3}$
Thermal expansivity	$3.5 \times 10^{-5}$	$\alpha$	$\text{K}^{-1}$

**Table S1.** Parameters used in the reference model. See also the included ASPECT input file (`input_reference_model.prm`). The alternative model differs only in that  $\gamma_0^\phi = \gamma_0^C = 2$ .

**129 Movie S1 and S2 Captions**

130 **Movie S1** shows evolution of the reference model. The top panel shows the hor-  
131 izontal component ( $D_{xx}$ ) of the strain rate tensor for the reference model, at times la-  
132 belled. The model velocity field is shown with arrows. The bottom panel shows the vor-  
133 ticity, along with the accumulated plastic strain in greyscale, saturated at a value of 0.7.  
134 Bottom panel also shows vectors of the translated velocity field (velocity in the hang-  
135 ing wall reference frame). The two black lines in each panel show contours of the tem-  
136 perature field at 600 and 700 °C.

137 **Movie S2** shows evolution of alternative model, where the strain intervals that de-  
138 termine plastic strength weakening are reduced. All features shown are identical to Movie  
139 S1.

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